

H. O. Fuchs¹

Approximate Analysis for Optimizing Prestress Treatments

REFERENCE: Fuchs, H. O., "Approximate Analysis for Optimizing Prestress Treatments," *Analytical and Experimental Methods for Residual Stress Effects in Fatigue*, ASTM STP 1004, R. L. Champoux, J. H. Underwood, and J. A. Kapp, Eds., American Society for Testing and Materials, Philadelphia, 1988, pp. 13-20.

ABSTRACT: A method of analysis for finding optimum shot peening specifications for given parts and loading conditions is explained. The method is applied to a part with fillets loaded in rotating bending. The computed optimum intensity is 9 C. Fatigue tests are reported which compare this intensity to the conventionally specified 7A. The higher intensity proved clearly superior.

KEY WORDS: shot peening, optimization, crack arrest, stress profiles, fatigue tests

The great improvement in fatigue strength that can be obtained by prestress treatments such as shot peening has been known for about 50 years [1]. When production quantities were large enough to justify the expense of repeated fatigue tests the treatments have been optimized empirically. For other applications certain rules have been established that recommend or prescribe the treatment, for instance, the specification for shot peening Military Specification for Shot Peening of Metal Parts (MIL-S-13165).

This paper shows a method for finding optimum treatments analytically and reports tests that validated the method by comparing conventional peening to optimized peening, which was four times as heavy as the conventional treatment.

Criteria and Assumptions

Failures start on the surface in parts that are free of self-equilibrating stresses (called self stresses or residual stresses) and of internal defects. The surface is weaker than the interior because it is exposed to corrosion, it has some roughness, and its particles are not supported by as many neighboring particles as those on the interior. Also, and more importantly, there usually is a stress gradient, with stresses decreasing going from the surface inwards. We neglect the surface weakness and consider only the effects of the stress gradients because shot peening is not very effective in improving fatigue strength in pure axial loading, but is more effective in providing this improvement under the bending and torsion of smooth parts, and surprisingly effective for notches that have high stress gradients.

Producing compressive self stresses in the skin is the main effect of mechanical prestress treatments. We neglect other effects such as changes in structure produced by peening. The compressive stresses in the skin are balanced by tensile stresses in the core. The compressive stresses increase fatigue resistance; the tensile stresses decrease it. Greater peening intensity decreases the risk of fatigue failures that start from the surface and increases the risk of fatigue

¹Professor, Mechanical Engineering Department, Stanford University, Stanford, CA 94305.

... extending below the surface. Equality of risks of failures that start from the surface or from the interior is our criterion for optimum peening intensity.

Fatigue failure of a part is expected if the stresses at some point exceed the critical stresses for crack initiation and the resulting crack is not arrested. The critical stress for initiation depends on the material and on the mean stress. The critical stress for crack propagation depends on the threshold stress intensity factor. We assume that factor to be equal to zero ksi in.^{1/2} or MPa · m^{1/2}; this is equivalent to assuming that cracks cannot propagate unless the maximum stress is tensile.

To check the stresses at different points against those criteria, we need to know the profiles, the distribution in depth of the load stresses and of the self stresses. The load stress profile at fillets is not readily available. We compared the finite element results produced by Tipton [2] with formulas used to arrive at the stress gradient at the surface developed by Siebel and Meuth [3] and developed a formula

$$L = (ax + bx^n)S \quad (1)$$

where S is the nominal bending stress at the surface, L is the load stress at point x , x is the distance from the center divided by the radius, and a , b and the exponent n are constants that satisfy the stress concentration at the surface, the gradient at the surface, and the equilibrium with external bending moment.

Figure 1 shows the profile for a 0.035-in. (0.9-mm) fillet radius between the diameters of 1 and 0.8 in. (25 and 20 mm).

For the self stress profile, we assumed that the maximum self stress produced by peening depends only on the material, not on the intensity, provided the shot is harder than the material. We took the depth of compressive stress and the maximum from data by Brodrick [4], and assumed that the maximum always occurs at 25% of the depth, and that the profile is parabolic for the compressive self stress. We assumed that the tensile self stress is uniformly distributed in

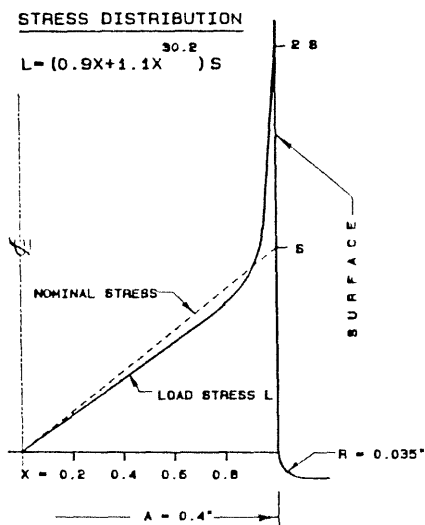


FIG. 1—Profile of load stress at the fillet of the test specimens.

the core of the round specimen. The concentration of the self stress at the notch [5] was neglected. Figure 2 shows the assumed profile.

We computed the decrease of the near surface self stress caused by yielding in the compressive part of the cycles, but did not change the rest of the profile. We also neglected the redistribution of stresses arising from the growth of circumferential cracks. We considered only the axial stresses; the circumferential and radial stresses below the notch were neglected.

Computation

The fatigue limit for the specimen shown in Fig. 3 was calculated for optimum shot peening and for shot peening according to MIL-S-13165. The calculated optimum intensity was 9 C, which is more than four times the conventional intensity of 7A. The calculated fatigue limit with 9 C was 23% higher than it was with 7A.

The computations were done on an IBM PC computer. For the given material properties and stress distributions, a bending moment was assumed. Stresses at small increments of depth were computed and compared with failure criteria. If there was no crack initiation, or if there was crack arrest at a point below the initiation, the bending moment was increased and the process repeated until a crack was initiated and not arrested. The bending moment previous to this last one was considered the calculated fatigue limit.

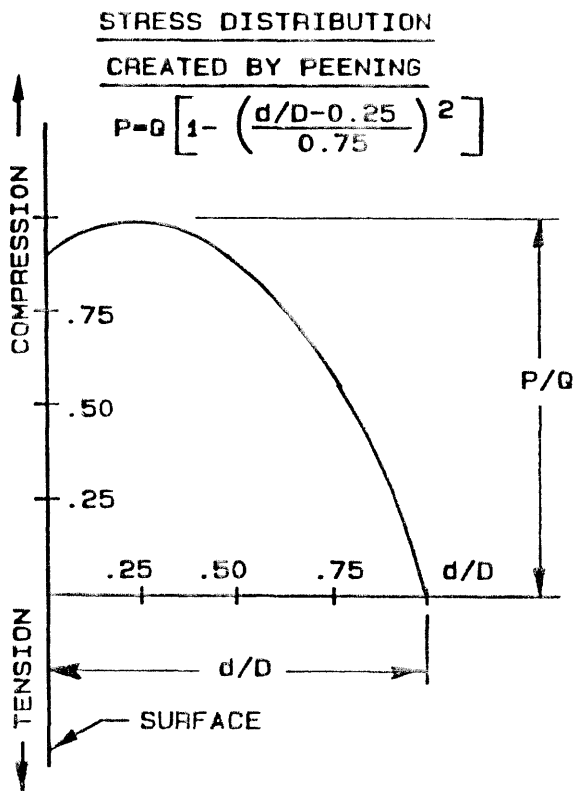


FIG. 2—Profile of self stresses created by peening.

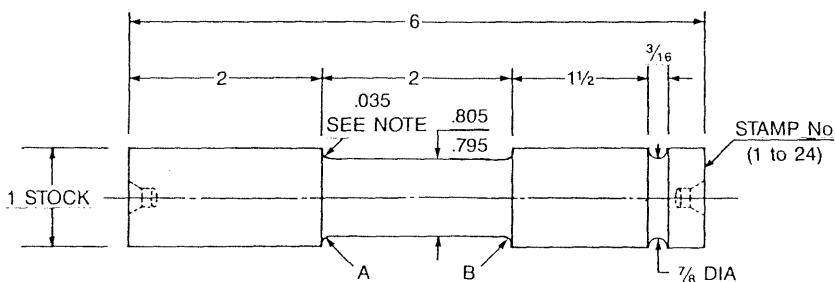


FIG. 3—Peened specimens fatigue tested at lives of about 75,000 cycles; material: 4340 steel; (1) 0.035 radius ± 0.005 , but radii at A and B must be equal within ± 0.002 , (2) diameters at A and B must be equal within ± 0.002 ; no undercut or step, (3) after finish machining, heat in neutral atmosphere, quench in oil, temper to $R_c 52$; do not straighten, (4) shot peen per 86.02.26, (5) check hardness on 1-in. diameter.

The following material properties were used in the calculations:

- cyclic yield strength, 1600 MPa (230 ksi);
- fatigue limit, 556 MPa (80 ksi);
- mean stress coefficient,² -0.35 ; and
- maximum compressive self stress, 883 MPa (127 ksi);

The calculated fatigue limits were

- 334 MPa (48 ksi) with 0.005 in. (0.13-mm) depth of compression (7A intensity) and;
- 410 MPa (59 ksi) with 0.015 in. (0.38-mm) depth of compression (9C intensity).

Experiments

Fatigue tests were conducted to validate the calculations. They definitely established the greater strength of the fillets peened with the calculated optimum intensity as compared to fillets peened according to MIL-S-13165.

The tests, which are reported here, were only exploratory. They used a few specimens tested at lives of about 75 000 cycles to check whether the much greater effort required to establish quantitative data sets would be worth while. Figure 3 is a drawing of the specimens.

These tests were run at Union College, Schenectady, NY, by students of Prof. R. Eisenstadt. They used a rotating beam fatigue machine capable of over 565 N · m (5000 lb. in.) in four-point bending. For the first half of the test program, the machine needed frequent adjustments and improvements.

The first series of tests used six discriminating specimens [6]. The two fillets on each specimen are exposed to the same bending moment. One fillet was peened to 7A, the other to 9 C. The stronger fillet would remain unbroken when the weaker fillet failed. To make sure that accidental differences would not confuse the test, all specimens were mounted with the grooved end in the same position in the test machine. Three specimens had high peening intensity in the fillet near the grooved end, the other three had it in the other fillet. This strategy proved especially valuable because of the frequent revisions of the test machine; as both fillets were exposed to the same conditions, the results were conclusive in a qualitative sense, although somewhat doubtful in a quantitative sense. All six specimens failed in the fillet peened to 7A intensity, but none failed in the fillet peened to 9 C intensity.

²The mean stress coefficient is the slope of the line that plots alternating stress over mean stress at the fatigue limit.

To obtain a quantitative estimate of the degree of improvement, four more specimens were run in which both fillets had been peened to 9 C, and an attempt was made to get approximately the same life to failure.

The raw test data and computed data are shown in Table 1. A visual comparison is shown in Fig. 4. As the tests were run in steps with different bending moments, it was necessary to convert numbers of cycles to accumulated damage cycles. This was done by assuming that the damage cycles (n') were proportional to the ninth power of the respective bending moments.

Based on the logarithmic means of the bending moments and cumulative damage cycles, the strength improvement was about 4% at about 70,000 cycles. Based on the weakest of each set of specimens, the strength improvement was about 8% at about 35 000 cycles. In both sets of

TABLE 1—Step test results.

<i>M</i>	<i>n</i>	<i>n'</i>
1		
3 499	70 142	11 538
3 888	71 450	30 353
4 276	66 829	66 829
		108 710 = <i>N'</i>
2		
3 499	67 590	11 118
3 888	69 594	29 564
4 276	61 508	61 508
		102 190 = <i>N'</i>
3		
3 499	69 470	26 884
3 888	16 300	16 300
		33 184 = <i>N'</i>
4		
2 916	200 700	7 769
3 110	63 870	4 414
3 304	38 430	4 579
3 499	32 700	6 528
3 693	34 849	11 306
3 888	35 651	18 380
4 082	38 500	30 805
2 527	34 078	364
2 721	35 803	743
2 916	34 849	1 349
3 110	33 026	2 283
3 304	35 180	4 191
3 499	73 714	14 748
3 693	109 000	35 316
3 888	136 600	70 485
4 185	23 574	23 574
		236 834 = <i>N'</i>
5		
3 499	72 881	11 989
3 888	73 889	31 389
4 276	27 126	27 126
3 797	13 193	4 529
		75 033 = <i>N'</i>

TABLE 1—Continued.

<i>M</i>	<i>n</i>	<i>n</i> '
	6	
3 499	47 228	9 428
3 888	57 711	29 753
4 185	19 068	19 068
		58 249 = <i>N</i> '
	7	
4 573	35 668	35 668 = <i>N</i>
	8	
4 573	49 379	49 379 = <i>N</i> '
	9	
4 185	32 604	32 604 = <i>N</i> '
	10	
4 185	238 600	107 310
4 573	52 100	52 100
4 185	16 300	7 335
4 573	55 854	55 854
		222 597

NOTE: *M* = bending moment (lb. in.),
n = number of revolutions, cumulative cycles
n' = $n(M/M_{\max})^9$.

NOTE: log mean for #1 to #6:

<i>M</i>	3.631	<i>n</i> '	5.036
	3.631		5.009
	3.590		4.521
	3.622		5.374
	3.631		4.875
	3.622		4.765
($\delta = 0.015$)	3.621	($\delta = 0.287$)	4.930

mean *M* = 4 178 lb. in.
mean *N*' = 85 100 cycles

log mean for #7 to #10

<i>M</i>	3.660	<i>n</i> '	4.746
	3.660		4.694
	3.622		4.513
	3.660		5.348
($\delta = 0.019$)	3.651	($\delta = 0.362$)	4.825

mean *M* = 4 472 lb. in.
mean *N*' = 66 800 cycles

specimens, there was a weak specimen that had the shortest life and also the lowest bending moment at failure.

Discussion

The mechanism by which compressive self stresses arrest cracks that start from notches has been investigated by Heller et al. [7] and by Gerber and Fuchs [8]. They looked at the stres

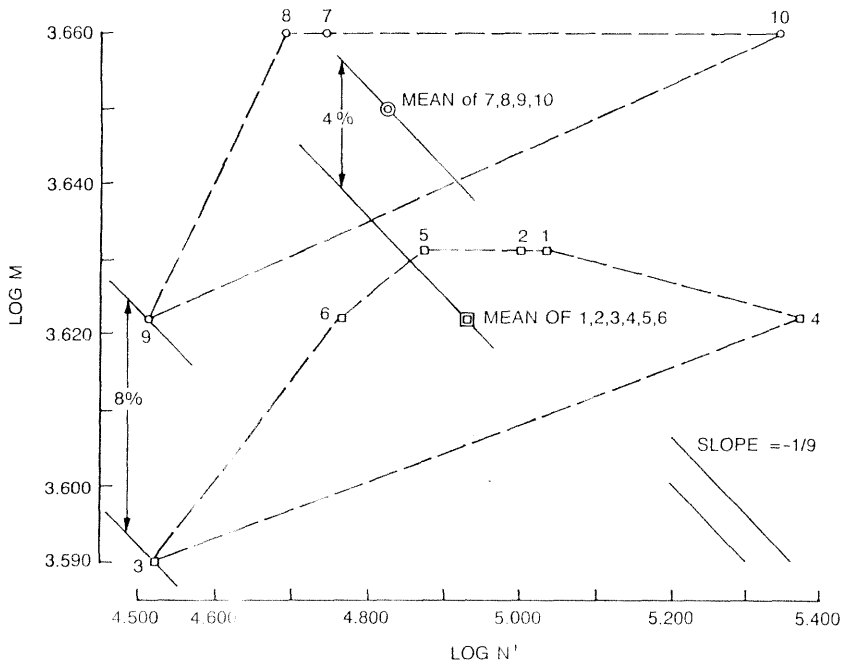


FIG. 4—Fatigue test results.

profile below V-grooves for which approximate solutions had been developed by Neuber [9]. The present work looks at fillets for which no general solutions are known. It simplifies the analysis by neglecting a number of factors. The way in which shot peening raises fatigue strength by arresting cracks at some depth below the surface of smooth bending specimens was shown by Starker et al. [10].

The use of cumulative damage cycles in evaluating the results of stepped fatigue tests requires an assumption about the unknown S-N curve. We assumed a fatigue strength exponent equal to -0.11 . This compares with exponents of about -0.09 given in the literature for similar steels. We assumed that the notched, peened part would have a stronger slope than the smooth specimens would. The stronger slope favors the conventionally peened specimens in the comparison.

Conclusion

A simplified analysis showed that the optimum shot peening intensity for a fillet is more than four times the conventionally specified intensity. Exploratory tests showed that the computed optimum intensity is definitely superior at fatigue lives between 30,000 and 200,000 cycles.

Acknowledgments

This work was sponsored by the Metal Improvement Company, which provided financial support and advice about shot peening. The computer program was developed by Saeed Benaie. The tests were carried out under the supervision of Professor R. Eisenstadt of Union College. The author is grateful for this help.

References

- [1] Almen, J. O., *Residual Stresses and Fatigue in Metals*, McGraw-Hill, New York, 1963, p. 64.
- [2] Tipton, S. M., Fatigue behavior under multiaxial loading in the presence of a notch, Ph.D. Dissertation, Stanford University, Stanford, CA, 1984, p. 134.
- [3] Siebel, E. and Meuth, H. O., "Die Wirkung von Kerben bei schwingender Beanspruchung," *VDI Zeitschrift*, Vol. 91, 1949, pp. 319-323.
- [4] Brodrick, R. F., "Protective shot peening of propellers," WADC Technical Report 55-56, Part I, p. 25-36. Also shown in "Shot Peening Applications," Metal Improvement Co., Paramus, N.J., 1980, p. 10.
- [5] Todd, R. H. and Fuchs, H. O., "Self stress concentrations," *Experimental Mechanics*, Vol. 11, 1971, pp. 548-553.
- [6] Fuchs, H. O., "Fatigue research with discriminating specimens," *Fatigue of Engineering Materials and Structures*, Vol. 2, 1979, pp. 207-215.
- [7] Heller, R. A., Seki, M., and Freudenthal, A. M., "The effects of residual stress on random fatigue life," *Proceedings of ASTM*, Vol. 64, pp. 516-535.
- [8] Gerber, T. L. and Fuchs, H. O., "Improvement in the fatigue strength of notched bars by compressive self stresses," in *Achievement of High Fatigue Resistance in Metal and Alloys STP 467*, American Society for Testing and Materials, Philadelphia, 1970, pp. 276-295.
- [9] Neuber, H., *Kerbspannungslehre*, Springer Verlag, Berlin, 1958.
- [10] Starker, P., Wohlfahrt, H., and Macherauch, E. "Biegewechselfestigkeit und Groesseffekt bei unterschiedlich warmbehandelten Stahlproben aus CK 45 nach Kugelstrahlen" *Proceedings of the First International Conference on Shot Peening*, Pergamon, Oxford, 1981, pp. 613-623.

References

- [1] Almen, J. O., *Residual Stresses and Fatigue*
- [2] Tipton, S. M., Fatigue behavior under motion, Stanford University, Stanford, CA,
- [3] Siebel, E. and Meuth, H. O., "Die Wirbel", *Zeitschrift*, Vol. 91, 1949, pp. 319-323.
- [4] Brodrick, R. F., "Protective shot peening", pp. 25-36. Also shown in "Shot Peening Applications", p. 10.
- [5] Todd, R. H. and Fuchs, H. O., "Self stresses", pp. 548-553.
- [6] Fuchs, H. O., "Fatigue research with disordered structures", Vol. 2, 1979, pp. 207-211.
- [7] Heller, R. A., Seki, M., and Freudenthal, "Fatigue life", *Proceedings of ASTM*, Vol. 64, pp. 1-10.
- [8] Gerber, T. L. and Fuchs, H. O., "Improving self stresses," in *Achievement of High Strength*, Society for Testing and Materials, Philadelphia.
- [9] Neuber, H., *Kerbspannungslehre*, Springer.
- [10] Starker, P., Wohlfahrt, H., and Macher, "The effect of differently heat-treated steel specimens on shot peening", *International Conference on Shot Peening*.